# AI-As-Ni (Aluminum-Arsenic-Nickel)

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The previous work on this ternary system is limited to a computed isothermal section at 25 °C by [1989Kli]. More recently, [1995Dep] determined an isothermal section at 800 °C, which depicts three ternary phases.

## **Binary Systems**

The Al-As phase diagram [1984Mca] has one stoichiometric compound AlAs (*B*3, ZnS (sphalerite)-type cubic), which melts congruently at ~1760 °C. The Al-Ni phase diagram [1993Oka] shows five intermediate phases: NiAl<sub>3</sub> (*D*0<sub>11</sub>, Fe<sub>3</sub>C-type orthorhombic), Ni<sub>2</sub>Al<sub>3</sub> (*D*5<sub>13</sub>-type hexagonal), NiAl (*B*2, CsCl-type cubic, also denoted  $\beta$ ), Ni<sub>5</sub>Al<sub>3</sub> (Ga<sub>3</sub>Pt<sub>5</sub>-type orthorhombic), and Ni<sub>3</sub>Al (*L*1<sub>2</sub>, AuCu<sub>3</sub>-type cubic; also denoted  $\gamma'$ ). The As-Ni phase diagram [1987Sin] has five intermediate phases: Ni<sub>5</sub>As<sub>2</sub> (hexagonal), Ni<sub>11</sub>As<sub>8</sub> (tetragonal), NiAs (*B*8<sub>1</sub>-type hexagonal), and two modifications of NiAs<sub>2</sub> with the transition temperature at ~600 °C. αNiAs<sub>2</sub> is orthorhombic (space group *Pbca*), and  $\beta$ NiAs<sub>2</sub> is a *C*18, FeS<sub>2</sub> (marcasite)-type orthorhombic phase.

## **Ternary Phases**

[1989Kli] did not include any ternary phases in their computation of the phase equilibria of this system at 25 °C.

[1995Dep] found three ternary phases in samples annealed at 800 °C and quenched in water. All three phases have hexagonal structures derived from the NiAs-type structure. Two of the phases denoted A and D, respectively, by [1995Dep] have fully disordered structure with the lattice parameters comparable to those of NiAs, whereas the B phase is a superstructure with  $a = a' \sqrt{3}$  and c = 3c', where a' and c' are the parameters of the NiAs subcell. The A phase has a composition range of Ni<sub>2.2-3</sub>Al<sub>0.35-0.6</sub>As<sub>1.4-1.65</sub> with a = 0.3685 to 0.3836 nm and c = 0.5035 to 0.5115 nm. The D phase has a range of Ni<sub>3.5</sub>Al<sub>0.8-1.5</sub>As<sub>0.5-1.2</sub> with a =0.3937 to 0.3973 nm and c = 0.5089 to 0.5027 nm. The B phase has a composition range of Ni<sub>3</sub>Al<sub>0.5-0.85</sub>As<sub>1.5-1.15</sub> with a = 0.6661 to 0.6687 nm and c = 1.5321 to 1.5318 nm.

#### **Isothermal Section**

With starting materials of purity  $\geq 99.9\%$ , [1995Dep] arc melted about 20 ternary alloys in Ar atm. The alloy samples were given a final anneal at 800 °C for 20 days and quenched in water. The phase equilibria were studied by x-ray powder diffraction and electron probe microanalysis. The compositions of the coexisting phases were listed. The isothermal section constructed by [1995Dep] at 800 °C is redrawn in Fig. 1. The Ni-Al and Ni-As binary compounds



Fig. 1 Al-As-Ni isothermal section at 800 °C [1995Dep]

show very limited solubility for the third component, with the exception of NiAs, which dissolves Al up to the composition  $NiAs_{0.7}Al_{0.3}$ . The three ternary phases and (Ni) do not form tie-lines with AlAs. The phases NiAl,  $Ni_2Al_3$ , and NiAs form tie-lines with AlAs.

[1995Dep] also determined an approximate isothermal section at 800 °C for the Al-As-Ga-Ni quaternary system at the constant composition of  $Ga_{0.7}As_{0.3}$  and compared it with the section for this ternary system as well as that for the As-Ga-Ni system [1989Gue].

#### References

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